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Thermally stimulated conductivity in InGaAs/GaAs quantum wire heterostructures

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Abstract. Thermally stimulated conductivity of the InGaAs-GaAs heterostructures with quantum wires was studied using different quantum energies of exciting illumination. The structures reveal long-term photoconductivity decay within the temperature range 100 to 200 K, and effect of residual conductivity after turning-off the illumination. Analyzing the data of thermally stimulated conductivity, the following energies of electron traps were found: 90, 140, and 317 meV. The role of deep traps in recombination process as well as the photoconductivity mechanism was discussed.

Keywords: heterostructure, quantum wire, photoconductivity.

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1. Introduction

Nowadays the interest in studying the physical properties of heterostructures InGaAs-GaAs, specifically centers with deep and shallow trap levels, has increased significantly [4-7]. It is caused by its successful use to solve the scientific and technical problems for several years. InGaAs-GaAs heterostructure with quantum wires (QWR) are known for their unique photoelectrical properties that are perspective for implementation of new phototransistor technologies, infrared photo-detectors and solar cells.

Like to other semiconductor materials, deep defect states in the band gap of InGaAs are formed by doping with a wide class of impurities [5]. These states may be significantly complicated due to formation of dopant pairs or their interaction with their own lattice defects. Optical and electronic properties are studied by photocurrent (PC) spectra and thermal activation effects measurement [3]. However, thermally stimulated conductivity (TSC) spectra of InGaAs-GaAs heterostructures with QWR are not examined enough. According to [1-3], distribution of localized states in semiconductors is based on the multiple trapping model. It was shown that peaks in the spectra TSC were observed within the temperature range 100...150 K.

The main goal of this work is to obtain TSC spectra for heterostructures InGaAs-GaAs and to interprete them. To reach the goal, the classic method for obtaining TSC spectra and periodic light excitation method during the heating from 83 to 276 K were used in this paper.

2. Experimental details

All heterostructures were grown on GaAs (311)A semiinsulating substrates by using molecular beam epitaxy. After growing the 0.5- μ m GaAs buffer layer at 580 °C, the substrate temperature was reduced down to 520 °C for deposition of the InGaAs and *n*-type GaAs barriers. Five periods of InGaAs/GaAs were deposited with 6, 8, 10 and 11 monolayers (MLs) of In_{0.38}Ga_{0.62}As. Each GaAs barrier was 40-nm thick consisting of 10-nm

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undoped GaAs, then 20-nm uniformly doped GaAs:Si $(N_d = 5 \cdot 10^{17} \text{ cm}^{-3})$, and then another 10-nm undoped GaAs. Finally, each sample was capped with a final InGaAs QWR layer of equal deposition to the buried QWRs for AFM study. Samples with 11 and 10 MLs of InGaAs deposition are shown to form high-quality QWRs [7], while structures with 6 and 8 MLs maintain quantum well-like nature.

In order to get results, firstly, we cooled the sample to the temperature of 83 K, then linearly heated and periodically lighted samples (for 30 s). Spectra of lateral PC for 11 ML InGaAs-GaAs QWR hetero-structure measured within the range 0.6...2.0 eV at the temperature 83 K. The time dependence of PC rise and decay obtained during excitation by using light with energies $hv_1 = 1.35$ eV and $hv_2 = 1.65$ eV within the temperature range 83 to 296 K.

3. Results and discussion

Fig. 1 shows the time dependence of photoconductivity damping and decay, measured at different temperatures for InGaAs-GaAs structure with light excitation $hv_1 = 1.35 \text{ eV}$ and $hv_2 = 1.65 \text{ eV}$. This experiment was performed under the condition of a linear temperature increase at the speed close to 0.05 K/s, and therefore, a change in temperature during the relaxation process is negligible (see Fig. 2a, insert). The kinetics of PC decay was described by a stretched exponential dependence:

$$I_{PC} \sim \exp\left\{-\left(t/\tau\right)^{\beta}\right\},\tag{1}$$

where τ is the time constant, β – ideality factor, which is equal to $\beta = 0.7$ for the studied structures. Note that the law proved to an exponential PC increase, and the ratio of time decay constant to the time increase one was $\tau_{dec}/\tau_{rise} = 2.52$. It means that in addition to

recombination centers, long term electronic traps (shallow trap centers) have a significant impact on PC decay [4, 6].



Fig. 1. Growth diagram of InGaAs-GaAs heterostructure.

Fig. 2a shows the extended relaxation (residual photoconductivity), when conductivity is not returned to its equilibrium value for a long (~30 s) time, under excitation $hv_1 = 1.35 \text{ eV}$ at temperatures between 120 and 150 K. We can distinguish two peaks of thermal conductivity in the dark background of conductivity (dashed curve), which is measured under similar experimental conditions, but without photoexcitation. Energy hv_1 and hv_2 were selected being based on the analysis of the spectral dependence of InGaAs-GaAs structure (Fig. 3), measured at 83 K. Photocurrent spectroscopy reveals several electron transitions indicating the complicated density of states spectrum in our samples (Fig. 3). Besides electron transitions between ground states of conduction and valence bands of InGaAs QWR (arrow 1), the PC signal below band gap of GaAs is the result of photoexcited electrons in both the buffer layer (arrow 2) and defect states of GaAs (arrows 4 and 5) [4]. Band-to-band absorption in GaAs is observed above 1.43 eV (arrow 3). The energy $hv_1 =$ 1.35 eV causes resonant excitation of InGaAs QWR as a result of band-to-band transitions, while $hv_2 = 1.65$ eV is also responsible for possible excitation of electron-hole pairs in GaAs. In addition, PC within the spectral range 0.7...1.3 eV is caused by transitions between GaAs deep states. These energies determined in [6] indicate the presence of deep levels in the system.

Fig. 4 shows the TSC curve for InGaAs-GaAs 11 ML structure, measured by heating after excitation $hv_1 = 1.35$ eV. We can select a wide peak of thermal conductivity, but in contrast to the kinetic method, we cannot distinguish the detail structure of the TSC spectrum.



Fig. 2. Time dependences of PC relaxation under light excitation hv_1 (a) and hv_2 (b) and various temperatures (83...276 K). (The dashed line corresponds to the dark current.)

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Fig. 3. Spectra of lateral PC for 11 ML InGaAs-GaAs QWR heterostructure measured at 83 K. The bias voltage 100 mV was applied along QWR ($\overline{233}$] direction).

TSC peak with a maximum at 160 K was observed within the temperature range from 130 up to 200 K (Fig. 4). Assuming that the recapture (capture other sticking centers) does not happen, we can estimate the depth of attachment center relative to the conduction band [5]:

$$\varepsilon_a = 23k_{\rm B}T_{\rm max}\,,\tag{2}$$

where $k_{\rm B}$ is the Boltzmann constant, $T_{\rm max}$ – temperature of the TSC peak. This formula is simplified and provides a relatively large error in assessing the activation energy. Using the equation (2), we can give rough estimate of the localization depth of charge carrier on a shallow center relatively to the center of the conduction band ε_a = 317 meV.



Fig. 4. Classical thermally stimulated conductivity for InGaAs-GaAs 11 ML heterostructure.

Considering the shallow centers that are surrounded by QWR, it is possible that they determine the heterostructure conductivity in non-equal state. Due to the fact that the studied heterostructure has trapping center for electrons, it can be argued that the excited QWR have negatively charged environment.

Temperature dependence of PC decay shown in Figs. 5a, 5b for $hv_1 = 1.35$ eV and 5c for $hv_2 = 1.65$ eV. We can see that in both cases two TSC peaks, but for hv_1 (Fig. 4c) the spectrum is more extended and shifted into the region of higher temperatures.

Fig. 5 shows that the activation energy for electrons at excitation hv_1 (1.35 eV) is equal to $\varepsilon_a = 229$ meV for the first maximum and to $\varepsilon_a = 337$ meV for the second one. The activation energy at excitation hv_2 (1.65 eV) is equal to $\varepsilon_a = 218$ meV for the first maximum and to $\varepsilon_a = 278$ meV for the second maximum.



Fig. 5. (a) Dependence of the TSC spectra on heating (excitation $hv_1 = 1.35$ eV). Thermally stimulated conductivity of the sample 11 ML InGaAs-GaAs for various excitation intensities, (b) for $hv_1 = 1.35$ eV, (c) for $hv_2 = 1.65$ eV.

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Fig. 6. Energy band diagram of the 11 ML $In_xGa_{1-x}As$ structure. Where x = 0.38, QWR height is 3.8 nm.

When we use the formula (2) for assessing the activation energy, we obtain a sufficiently large error, because the formula is simplified and ignores the multiple trapping. The resulting activation energy was much higher than the values obtained from the analysis of TSC kinetic method. It means that at stationary filling TSC revealed deep electronic states.

Fig. 6 shows a diagram of the energy levels in the 11 ML In_xGa_{1-x}As heterostructure, calculated using the software NextNano³ with the parameters x = 0.38 and QWR height of 3.8 nm. We revealed that the structure is characterized by potential variation caused by the intermediate layers of doped GaAs, which creates potential parabolic quantum wells (W = 101 meV). In addition, the system has quantum-sized states implied by the movement restriction of charge carriers in In_xGa_{1-x}As QWR. Thus, the system includes two subsystems of quantum-sized states due to the electron movement restriction in either GaAs buffer layer, or in In_xGa_{1-x}As QWR. According to band diagram calculations of the activation energy for GaAs and QWR, potential wells were equal to $\varepsilon_a = 87 \text{ meV}$ and $\varepsilon_a = 142 \text{ meV}$, respectively. Electrons are localized at a minimum of GaAs and can be spatially separated from the holes. These photoelectrons cause extended photocurrent relaxation, because for their recombination with holes in In_xGa_{1-x}As it is necessary to overcome the potential barrier height $\varepsilon_a^{\text{GaAs}} = 87 \text{ meV}$ (see Fig. 6).

It was found that the temperature dependence of the time constant τ within the temperature range between 80 and 150 K agrees well with Arrhenius behavior of decay, when $\ln \tau(T) \sim \varepsilon_a/kT$. After analyzing the dependence of τ on temperature for $hv_1 = 1.35$ eV and $hv_2 = 1.65$ eV, we obtained the activation energies $\varepsilon_a = 146$ meV and $\varepsilon_a = 92$ meV, respectively. With account of the importance of energy activation for potential wells in GaAs and QWR, obtained from the band energy diagram, we conclude that electrons mainly fill the quantum states of QWR under excitation hv_1 , and electrons fill GaAs potential well under excitation hv_2 .

Consequently, we get the photocurrent relaxation processes involving different energy states (recombination centers) when light excitation with different energies (1.35, 1.65 eV) is used.

4. Conclusions

Thermally stimulated conductivity of the InGaAs-GaAs heterostructures with quantum wires was studied using different quantum energies of exciting illumination. The spectrum of electronic states that determines recombination in InGaAs-GaAs heterostructures was obtained at different temperatures, using the TSC method and kinetic one with periodic illumination of the sample. The structures reveal long-term photoconductivity decay within the temperature range 100...200 K and effect of residual conductivity after turning-off the illumination. Analyzing the data of thermally stimulated conductivity, the following energies of electron traps were found: 90, 140, and 317 meV. The obtained activation energy for potential wells of the band structure and studied temperature dependence of the time constant τ show that light excitation with a selective energy for $In_xGa_{1-x}As$ QWR leads to increase in PC caused by electrons concentrated in QWR.

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