

Ellipsometric study of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals

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Abstract. Single crystals of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ ($x = 0, 0.25, 0.5, 0.75, 1$) solid solutions were grown by directional crystallization from the melt. Grown crystals were investigated using XRD and spectral ellipsometry. $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ crystals are characterized by formation of a continuous series of solid solutions with a cubic F-43m structure. The studied samples had a nonlinear spectral dependence of the refractive index n and the extinction coefficient k . The presence of a sharp maximum of n and a rather sharp decrease in the k spectral dependences are observed within the 570...730 nm range. The spectral dependence of the refractive index was described by the single-oscillator Wemple–DiDomenico (WD) model. Parameters of the Wemple–DiDomenico model for $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals were determined. The effect of heterovalent cationic $\text{P}^{5+} \rightarrow \text{Si}^{4+}$ substitution on parameters of the Wemple–DiDomenico model was discussed.

Keywords: argyrodite, single crystal, refractive index, band gap.

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1. Introduction

Ternary and quaternary silver-based argyrodites with the chemical formula $\text{Ag}_{12-x}\text{E}^{x+}\text{Chal}_{6-y}\text{Hal}_y$ ($\text{E}^{x+} - \text{Ga}^{3+}, \text{Si}^{4+}, \text{P}^{5+}$; $\text{Chal} - \text{S}^{2-}, \text{Se}^{2-}, \text{Te}^{2-}$; $\text{Hal} - \text{Cl}^-, \text{Br}^-, \text{I}^-$) have become notable materials in superionic and thermoelectric applications [1–5]. Recent studies have revealed that some ternary sulfides exhibit attractive photovoltaic and photocatalytic properties. It is reported that synthetic Ag_8SnS_6 in the form of nanocrystals or thin films has a band gap of $E = 1.4$ to 1.50 eV [6–8], while the band gap of the germanium-containing analogue Ag_8GeS_6 constitutes $1.4 \dots 1.466$ eV [9, 10]. Meanwhile, quaternary argyrodites have higher values of band gap: from 1.527 eV ($\text{Ag}_7\text{GeS}_5\text{I}$) to 2.030 eV ($\text{Ag}_6\text{PS}_5\text{I}$) [11, 12]. Thus, potentially the first one meets a theoretical requirement of the Shockley–Queisser limit [13] for light absorbers in simple solar cells and the second for tandem cells [14].

Quaternary argyrodites crystallize in a cubic crystal system, SG F-43m with a similar cell parameter [15, 16]. A characteristic feature of argyrodite is the presence of a significant number of partially occupied cation positions (Ag^+), causing a disorder of the cationic sublattice. As a result, the appearance of the so-called Urbach tails is observed for argyrodites. The temperature and compositional dependence of Urbach tails in the solid solution based on quaternary argyrodites are reported in [11, 12]. The refractive index investigations of quaternary argyro-

dites are limited to $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Ge}_x)\text{S}_5\text{I}$ and $\text{Ag}_7(\text{Si}_{1-x}\text{Ge}_x)\text{S}_5$. Also, these phases and their properties are the subject of theoretical calculations and molecular dynamics simulations [17–20] performed to better understand the influence of structure disordering on their properties.

This work aims to contribute to the optical characterization reported for single crystals of quaternary argyrodites by performing ellipsometry experiments for the first time.

2. Experimental

2.1. Sample preparation

The synthesis of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ alloys ($x = 0.25, 0.5, 0.75$) was carried out from previously synthesized quaternary $\text{Ag}_6\text{PS}_5\text{I}$ and $\text{Ag}_7\text{SiS}_5\text{I}$. Individual compounds were synthesized from high-purity elemental components of Ag (99.995%), P (99.9999%), or Si (99.9999%), S (99.9999%), and previously synthesized AgI, taken in appropriate proportions in quartz ampoules evacuated to 0.13 Pa according to the procedure described in Refs. [16, 21]. The $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ alloys ($x = 0.25, 0.5, 0.75$) were synthesized by a direct one-temperature method in quartz ampoules evacuated to 0.13 Pa. The $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ samples were heated at the rate 50 °C/h to a maximum temperature of 950 °C. At this temperature, the samples were in the melt for 72 h. Subsequently, the samples were cooled to room temperature at a similar rate. As a

result, bulk polycrystalline alloys $\text{Ag}_{6.25}\text{P}_{0.75}\text{Si}_{0.25}\text{S}_5\text{I}$, $\text{Ag}_{6.5}\text{P}_{0.5}\text{Si}_{0.5}\text{S}_5\text{I}$, and $\text{Ag}_{6.75}\text{P}_{0.25}\text{Si}_{0.75}\text{S}_5\text{I}$ weighing 20 g each were obtained.

To establish the technological regimes (melt and annealing zone temperatures) of single crystal growth (Table 1), the obtained alloys were studied using differential thermal analysis (Pt/PtRh thermocouples, heating/cooling rate 700 °C/h). With account of the nature of the temperature behavior of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ solid solutions, their single crystals were grown by the method of directional crystallization from melt-solution in a two-zone vertical resistance furnace. The single crystals were grown in evacuated quartz ampoules with a conical lower part. The single crystal growth regime included the following stages. (i) Formation of the nucleus in the lower conical part of the ampoule by the method of collective recrystallization for 48 h. (ii) The movement of the crystallization front at a rate of 0.5 mm/h. (iii) The annealing of the obtained single crystals for 72 h. (iv) The cooling of single crystals from the annealing temperature to room temperature at a rate of 5 °C/h. As a result, $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals of dark gray color with a metallic luster with a diameter of 1.2 cm and a length of 3 to 4 cm were obtained.

2.2. Methods

The $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ crystals were studied using X-ray diffraction (XRD) and spectral ellipsometry. For XRD analysis the AXRD Benchtop (Proto Manufacturing Limited) equipped with a DECTRIS MYTHEN2 R 1D detector, with Bragg–Brentano $\theta/2\theta$ imaging geometry, Ni filtered $\text{CuK}\alpha$ radiation, 10...60° 2θ angle scanning range with a dynamic region of interest and 1 s exposure was used. The grown crystals were ground in an agate mortar and sieved to obtain a homogeneous powder. Powder patterns were indexed and lattice parameters were calculated in the EXPO 2014 program [22].

The spectral ellipsometry analysis of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals was performed using the HORIBA Smart SE spectral ellipsometer, spectral range 440...1000 nm, beam angle 70°. The dispersion of the refractive index n and the extinction coefficient k were measured on plane-parallel single-crystal plates of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ solid solutions polished to optical quality with a thickness of 0.2 mm at the temperature of 25 °C. The results were analyzed using the DeltaPsi2 software with the well-known Tauc–Lorenz model [23], which allows the determination of some important optical parameters using the Δ and Ψ parameters obtained as a result of measurements.

Table 1. Technological regimes for growing $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals.

Composition	Melt zone temperature, °C	Annealing zone temperature, °C
$\text{Ag}_{6.25}\text{P}_{0.75}\text{Si}_{0.25}\text{S}_5\text{I}$	870	550
$\text{Ag}_{6.5}\text{P}_{0.5}\text{Si}_{0.5}\text{S}_5\text{I}$	910	570
$\text{Ag}_{6.75}\text{P}_{0.25}\text{Si}_{0.75}\text{S}_5\text{I}$	925	580

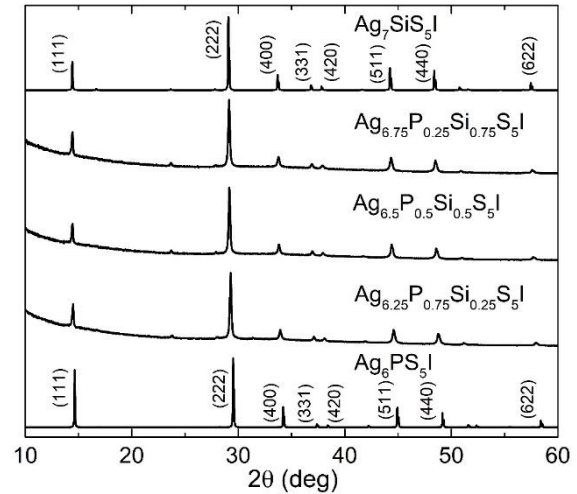


Fig. 1. Comparison of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ solid solutions powder patterns.

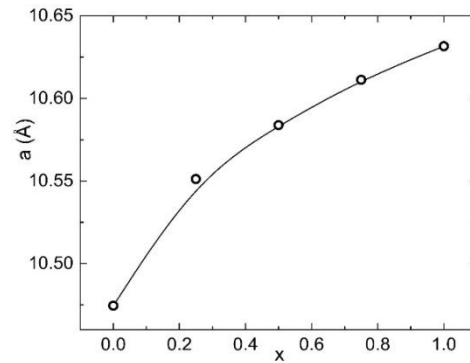


Fig. 2. Compositional dependence of lattice parameters for $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ solid solutions.

3. Results and discussion

As a result of comparison (Fig. 1) of the powder patterns of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ ($x = 0.25, 0.5, 0.75$) with those of $\text{Ag}_6\text{PS}_5\text{I}$ [16] and $\text{Ag}_7\text{SiS}_5\text{I}$ [21], the presence of one system of diffraction peaks was established, indicating the formation of a continuous series of solid solutions and the absence of impurity phases.

Thus, the single crystals of $\text{Ag}_{6.25}\text{P}_{0.75}\text{Si}_{0.25}\text{S}_5\text{I}$, $\text{Ag}_{6.5}\text{P}_{0.5}\text{Si}_{0.5}\text{S}_5\text{I}$ and $\text{Ag}_{6.75}\text{P}_{0.25}\text{Si}_{0.75}\text{S}_5\text{I}$ solid solutions, as well as individual $\text{Ag}_6\text{PS}_5\text{I}$ [16] and $\text{Ag}_7\text{SiS}_5\text{I}$ [21] crystallize in the structure of argyrodite in the space SG F-43m, $Z = 4$.

It was found that the compositional behavior of the lattice parameters (Fig. 2) has a monotonic nonlinear character (positive deviation from Vegard's rule), indicating the disorder of the crystal structures of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ solid solutions compared to $\text{Ag}_6\text{PS}_5\text{I}$ and $\text{Ag}_7\text{SiS}_5\text{I}$.

It has been established that the spectral dependences of the refractive index for all the compositions of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ solid solutions have a non-monotonic nonlinear character, which is expressed in the presence of a minimum and a maximum region (Fig. 3a). The presence of minimum regions (Fig. 3a) located within the

wavelength range of ~ 470 to 560 nm (depending on the composition of the solid solutions) is associated with interband optical transitions, while the maximum regions (~ 570 to 730 nm) are located within the optical absorption edge [24, 25]. It should be noted that within the transparency range, with a further increasing the wavelength the refractive index decreases monotonically for all solid solution compositions (Fig. 3a). It is noteworthy that, in contrast to the refractive index n , the spectral behavior of the extinction coefficient k is more monotonous (Fig. 3b), manifesting in the decrease in the values of extinction coefficient k with increasing wavelength (within the transparency range).

Thus, the heterovalent cationic $P^{+5} \rightarrow Si^{+4}$ substitution in $Ag_{6+x}(P_{1-x}Si_x)S_5I$ single crystals leads to a monotonic insignificant ($2.54 \dots 2.68$) increase in the refractive index n (Fig. 4) determined within the transparency range ($\lambda = 1000$ nm).

Let us consider the spectral dependence of the refractive index n using the single-oscillator Wemple–DiDomenico (WD) model in detail, where the variance of refractive index is described by the following relation [26]:

$$n^2(E) - 1 = \frac{E_d^{WD} E_0^{WD}}{(E_0^{WD})^2 - E^2}. \quad (1)$$

where E_0^{WD} is the energy of a single-oscillator, E_d^{WD} is the dispersion energy.

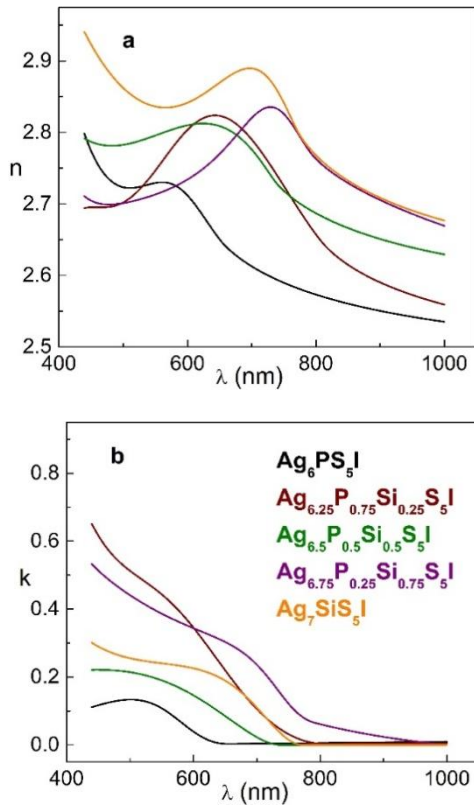


Fig. 3. Spectral dependences of refractive index n (a) and extinction coefficient k (b) for $Ag_{6+x}(P_{1-x}Si_x)S_5I$ crystals.

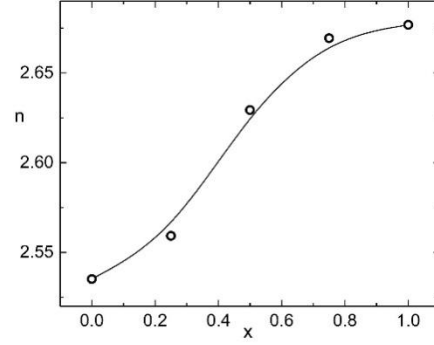


Fig. 4. Compositional dependence of the refractive index n for $Ag_{6+x}(P_{1-x}Si_x)S_5I$ single crystals.

The energy of a single-oscillator determines the spectral variation of the refractive index and describes the average energy of electrons transition between the highest energy levels of the valence band top and the lower levels of the conduction band in the band structure. Thus, using the WD model, the band gap E_g can be estimated as

$$E_0^{WD} \approx 2E_g^{WD}. \quad (2)$$

It should be noted that the dispersion energy reflects the chemical nature and structure of the substance and characterizes the average strength of the interband optical transitions, which is shown by an empirical relation [27]:

$$E_d^{WD} = \beta N_c Z_a N_e, \quad (3)$$

where β is a constant, typically $\beta = 0.37 \pm 0.04$ eV for compounds with a covalent bonds and $\beta = 0.26 \pm 0.04$ eV for compounds with an ionic bonds, N_c is the coordination number of the cation in the anionic polyhedron, N_e is the total number of valence electrons per anion, Z_a is the anion oxidation state.

Considering the above equations, one can summarize that the dispersion energy is associated with changes in the structural order of the material (the degree of ionic bonding, anion's oxidation state, the coordination number in the structural polyhedra, etc.) [28, 29].

The values of the static (constant at an infinite wavelength) refractive index n_0^{WD} and ionicity f_i^{WD} were calculated within the Wemple–DiDomenico model by using equations (4) and (5), respectively [29]:

Table 2. Parameters of the Wemple–DiDomenico model for $Ag_{6+x}(P_{1-x}Si_x)S_5I$ single crystals.

Composition	E_0^{WD} (eV)	E_d^{WD} (eV)	E_g^{WD} (eV)	n_0^{WD}	f_i^{WD}
Ag_6PS_5I	5.17	26.49	2.59	2.47	0.44
$Ag_{6.25}P_{0.75}Si_{0.25}S_5I$	4.85	25.46	2.43	2.50	0.44
$Ag_{6.5}P_{0.5}Si_{0.5}S_5I$	4.38	23.80	2.19	2.53	0.43
$Ag_{6.75}P_{0.25}Si_{0.75}S_5I$	3.88	21.39	1.94	2.55	0.42
Ag_7SiS_5I	3.72	20.60	1.86	2.56	0.42

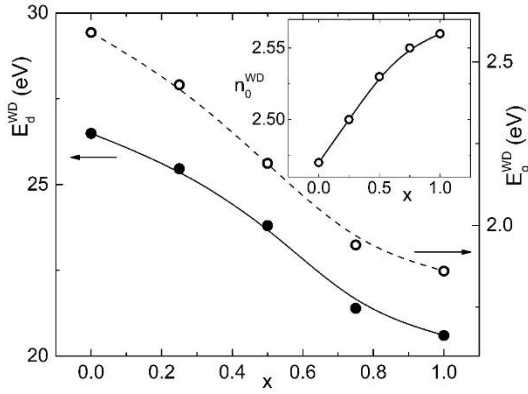


Fig. 5. Compositional dependences of the dispersion energy E_d^{WD} and the estimated band gap values E_g^{WD} for $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals. The inset shows the compositional dependence of the static refractive index n_0^{WD} .

$$n_0^{\text{WD}} = \left[1 + \frac{E_d^{\text{WD}}}{E_0^{\text{WD}}} \right]^{1/2}, \quad (4)$$

$$f_i^{\text{WD}} = \left[\frac{E_0^{\text{WD}}}{E_d^{\text{WD}}} \right]^{1/2}. \quad (5)$$

All the parameters calculated within the Wemple–DiDomenico model for $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ single crystals are shown in Table 2.

Thus, it was found that the heterovalent cationic $\text{P}^{+5} \rightarrow \text{Si}^{+4}$ substitution leads to a monotonic nonlinear decrease in the values of E_0^{WD} (Table 2), E_d^{WD} , and E_g^{WD} (Fig. 5), and an increase in the values of the static refractive index n_0^{WD} (inset of Fig. 5), while the values of ionicity f_i^{WD} practically do not change (Table 2).

The decrease in the energy of a single-oscillator E_0^{WD} (Table 2) and, as a consequence, the band gap values E_g^{WD} (Fig. 5) are associated with the changes in $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ band structure. Since, the optical parameters of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ are mainly responsible for the tetrahedra $[\text{PS}_4]$ ($\text{Ag}_6\text{PS}_5\text{I}$), $[\text{SiS}_4]$ ($\text{Ag}_8\text{SiS}_6\text{I}$) and $[\text{P}(\text{Si})\text{S}_4]$ ($\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ ($x = 0.25; 0.5; 0.75$)), which emerging from (Eq. (3)) and the results of calculating ionicity f_i^{WD} (Table 2), one can summarize the following. The decrease in the values of dispersion energy E_d^{WD} (Fig. 5) induced by the heterovalent cationic $\text{P}^{+5} \rightarrow \text{Si}^{+4}$ substitution is associated with a gradual decrease in the number of valence electrons N_e from 11 ($[\text{PS}_4]$ tetrahedron) to 10 ($[\text{SiS}_4]$ tetrahedron) since the other parameters (Eq. (3)) for the above tetrahedra will not change.

4. Conclusions

Grown using the directional crystallization $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ ($x = 0, 0.25, 0.50, 0.75, 1$) crystals were studied using the powder XRD and spectral ellipsometry methods. It has been ascertained that all studied samples crystallize in a cubic SG F-43m, indicating the formation of a continuous series of solid solutions. The lattice parameters of $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ were determined. The spectral dependences of the refractive index n and the

extinction coefficient k were investigated. The obtained results were analyzed within the Tauc–Lorenz model. To describe the dependences of the refractive index and calculate the single-oscillator E_0^{WD} and dispersion E_d^{WD} energies, the Wemple–DiDomenico single-oscillator model was used. It was ascertained that the heterovalent cationic $\text{P}^{+5} \rightarrow \text{Si}^{+4}$ substitution induces the monotonic nonlinear decrease in the compositional dependences of E_0^{WD} and E_d^{WD} .

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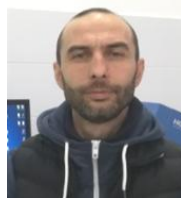
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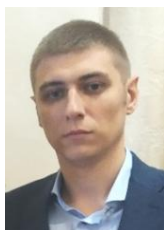
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Еліпсометричне дослідження монокристалів $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$

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Анотація. Монокристали твердих розчинів $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ ($x = 0, 0.25, 0.5, 0.75, 1$) вирощували методом спрямованої кристалізації з розплаву. Вирощені кристали досліджували методами рентгенівського фазового аналізу та спектральної еліпсометрії. Кристали $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$ характеризуються утворенням неперервного ряду твердих розчинів з кубічною структурою F-43m. Досліджувані зразки мають нелінійну спектральну залежність показника заломлення n і коефіцієнта екстинкції k . В області 570...730 нм спостерігається наявність різкого максимуму для n і досить різкий спад спектральних залежностей k . Спектральна залежність показника заломлення описується одно-осциляторною моделлю Вемпла–ДіДоменіко. Визначено параметри моделі Вемпла–ДіДоменіко для монокристалів $\text{Ag}_{6+x}(\text{P}_{1-x}\text{Si}_x)\text{S}_5\text{I}$. Обговорено вплив гетеровалентного катіонного заміщення $\text{P}^{+5} \rightarrow \text{Si}^{+4}$ на параметри моделі Вемпла–ДіДоменіко.

Ключові слова: аргіродит, монокристал, показник заломлення, заборонена зона.