Localized surface plasmon resonance in Au nanoprisms on glass substrates

O.G. Lopatynska¹, A.M. Lopatynskyi², T.I. Borodinova³, V.I. Chegel², L.V. Poperenko¹
¹Taras Shevchenko National University of Kyiv, 64/13, Volodymyrska Str., 01601 Kyiv, Ukraine
²V. Lashkaryov Institute of Semiconductor Physics, NAS of Ukraine, 41, prospect Nauki Ave., 03680 Kyiv, Ukraine
³F. Ovcharenko Institute of Biocolloid Chemistry, NAS of Ukraine, 42, Acad. Vernadsky Blvd., 03680 Kyiv, Ukraine
Phone: +38 (050) 824-42-03, e-mail: olga_lopatynska@ukr.net, lop2000@ukr.net, borodinova@ua.fm, vche111@yahoo.com, plv@univ.kiev.ua

Abstract. Metal nanocrystals are actual objects for the modern biophysics mainly because of their usage in sensors based on localized surface plasmon resonance (LSPR) and as active substrates for surface-enhanced spectroscopies. This work deals with the experimental and theoretical investigation of optical properties of trigonal and hexagonal Au nanoprisms deposited on the glass substrates. It was confirmed for the studied structures that the LSPR spectra depend on the crystals shape and size. Theoretical modeling the optical properties of plasmon-supporting nanoprisms was performed using the finite-difference time-domain method. The experimentally obtained and theoretically modeled LSPR spectral positions were found to be different, which can be attributed to a high spread of nanoprism shapes and sizes in the same sample and to nanocrystals aggregation effect confirmed by microscopy data. Additionally, the distributions of the electric field in the vicinity of nanoprisms under the LSPR conditions were simulated, and a strong field intensity enhancement at the corners of the prisms was demonstrated, which implies the promising application of such plasmonic nanostructures for surface-enhanced spectroscopy.

Keywords: localized surface plasmon resonance, Au nanopism, finite-difference time-domain method.

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1. Introduction

Localized surface plasmon resonance (LSPR) phenomenon plays a significant role in the modern nanobiophysics, especially, due to its promising applications in biosensing [1], surface-enhanced infrared absorption (SEIRA) [2] and surface-enhanced Raman spectroscopy (SERS) [3, 4]. It is known that the spectral position of LSPR and accompanying local electric field enhancement strongly depend on the shape and size of the nanoparticles [5], as well as on the interparticle distance and the level of ordering in the surface-bound nanoparticle arrays [6]. Therefore, depending on the plasmon resonance parameters needed, noble metal nanostructures of different sizes and morphologies are used as LSPR-supporting materials. For example, spherical Au nanoparticles of different sizes has been exploited for the response enhancement of the surface...
plasmon resonance sensing system, and the optimal nanoparticle diameter of 40 nm was reported to give the best amplification [7]. Henry et al. [8] investigated Ag nanocube ensembles in water and individual Ag nanocubes on the glass substrate and found striking differences in respective LSPR spectra due to an expressed multimodal LSPR excitation in the latter case. Size- and shape-dependent LSPR properties of noble metal nanodisks [9], nanorods [10] and nanopyramids [11] have been also demonstrated. From the theoretical point of view, optical properties of the plasmonic nanoparticles are studied by means of analytical and numerical methods to simulate interaction of electromagnetic waves with nanoparticles. One of the most common analytical techniques is Mie scattering approach [12], which is applicable only for the spherical or ellipsoidal particles, while for the nonspherical nanostructures numerical methods are usually preferred. One of them is the finite-difference time-domain method (FDTD) [13], which advantages and disadvantages have been reported earlier [14].

In this work, we experimentally investigate the LSPR properties of chemically synthesized trigonal and hexagonal Au nanoprisms deposited on the glass substrates and theoretically simulate LSPR spectra of these nanostructures by means of the finite-difference time-domain method. The analysis of local electric field distribution near the surface of trigonal and hexagonal Au nanoprisms with different lateral dimensions is also presented.

2. Microscopy of trigonal and hexagonal Au nanoprisms deposited on glass substrates

Fabrication technology of the studied samples was described in the previous work [15]. Briefly, after synthesis Au nanocrystals were separated from the disperse medium by centrifugation (centrifuge OPN-8, 800 min⁻¹, 60 min), rinsed with distilled water, precipitate was resuspended in the distilled water. The crystals suspension washed from the stabilizer was drop-casted on the glass substrates and dried at the room temperature. According to the protocol, plain Au nanoprisms have the shapes of regular triangle, hexagon and truncated triangle with the average linear size of 100 nm and height of 60 nm.

The microphotographs of the synthesized Au nanocrystals (Fig. 1a, b), dark field image of large Au crystal (Fig. 1c) and electron diffraction pattern of the latter crystal (Fig. 1d) confirming the nanoprism monocryallinity are presented in Fig. 1. The microphotographs were obtained using transmission electron microscope JEM 100CX (JEOL, Japan) with the accelerating voltage of \( U = 100 \) kV. Optical microscopy of the samples was performed using OLYMPUS GX-41 microscope (Melitek-Ukraine, Ukraine) with \( \times 100 \) objective in the light field mode. Optical microscopy revealed a huge spread of nanoprism sizes and the effect of nanocrystals aggregation (Fig. 1e, f).

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3. LSPR in the trigonal and hexagonal Au nanoparticles deposited on glass substrates

Light extinction spectra of the samples were measured using the LSPR spectrometer “Nanoplasmon-003” (V. Lashkaryov Institute of Semiconductor Physics, NAS of Ukraine).

Light absorbance measurements have shown the existence of a broad complex absorption band within the spectral range 500...1000 nm with a distinct peak in the visible region at 591 nm (Fig. 2). The appearance of this absorption band can be attributed to the LSPR excitation in surface-bound Au nanoparticles, while its complex form and broadening are possibly observed due to distribution of shapes and sizes of Au nanoparticles with several dominating geometrical configurations.

To investigate theoretically the LSPR spectra of trigonal and hexagonal Au nanoparticles, we modelled the light extinction cross-section spectra for trigonal Au nanoparticles with side dimensions of 100 nm (model 1, Fig. 3a), 25 nm (model 1, Fig. 3b), hexagonal Au nanoparticles with side dimensions of 33 nm (model 2, Fig. 3a) and 8 nm (model 2, Fig. 3b), which were obtained by truncation of trigonal nanoparticles to get regular hexagonal nanoparticle. Additionally, calculations were performed for the combination of trigonal Au nanoparticle with side dimension of 25 nm and hexagonal Au nanoparticle with side dimension of 8 nm with interparticle distance of 30 nm (model 3, Fig. 3b), which models the influence of nanoparticle shape distribution on the light extinction spectrum. In all the considered models, nanoparticles were located on glass substrates and the height of the nanoparticles was equal to 60 nm. All calculations were performed using the commercial FDTD package (Lumerical FDTD Solutions, trial version, Lumerical Solutions, Inc., Canada). The experimental absorbance spectrum and the theoretical extinction cross-section spectra corresponding to the abovementioned models are presented in Fig. 3.

As it can be seen from Fig. 3, simulated and experimental LSPR spectral positions and shapes are different. This can be explained by several factors related to the fact that theoretical spectra correspond to individual nanoparticles and experimental spectrum characterizes the whole sample. First, the interaction between the nanoparticles was not taken into account. Second, the effect of nanoparticles aggregation was present in the sample. Third, there were a broad distribution of nanoparticle sizes and several different nanoparticle shapes in the same sample. However, it should be noted that the best agreement with the experimental spectrum was produced by the model 2 (Fig. 3a). Therefore, we can conclude that the experimentally observed LSPR response results mainly from the 100 nm hexagonal Au nanoparticles, but the contribution of Au nanoparticles with other shapes and sizes is still significant.

Fig. 2. Experimental light absorbance spectrum of Au nanoparticles deposited on the glass substrate.

Fig. 3. The experimental absorbance spectrum of the Au nanoparticles on the glass substrate and the theoretical extinction cross-section spectra of (a) model 1 – trigonal Au nanoparticle with side dimension of 100 nm; model 2 – hexagonal Au nanoparticle with side dimension of 33 nm; (b) model 1 – trigonal Au nanoparticle with side dimension of 25 nm; model 2 – hexagonal Au nanoparticle with the side dimension of 8 nm; model 3 – combination of trigonal and hexagonal Au nanoparticles with side dimensions of 25 nm and 8 nm, respectively, and interparticle distance of 30 nm. All nanoparticles were located on the glass substrates and the height of nanoparticles was equal to 60 nm.
Fig. 5. The spatial electric field intensity distribution on the top bases of Au nanoprisms: (a), (b) trigonal nanoprism (model 1 in Fig. 3b); (c), (d) hexagonal nanoprism (model 2 in Fig. 3b); (e), (f) combination of trigonal and hexagonal nanoprisms (model 3 in Fig. 3b). All nanoprisms are located on the glass substrates and illuminated with orthogonal polarizations of the normally incident linearly-polarized light.

4. Electric field distribution in the vicinity of the Au nanoprisms on glass substrates

From the practical point of view, it is important to understand the relation between the geometrical parameters of plasmonic nanostructures and their performance as optical amplifiers, e.g., for application in surface-enhanced spectroscopy. To answer the question what shape and size of the abovementioned Au nanoprisms can provide higher signal enhancement, we calculated the electric field intensity distributions on the top bases of the investigated Au nanoprisms under the LSPR conditions using the FDTD method. Simulation results for the Au nanoprism models mentioned in Section 3 are presented in Figs 5 and 6. Calculations were performed for two orthogonal polarizations of the normally incident linearly-polarized light with the wavelength corresponding to the LSPR spectral position of the respective Au nanoprism model.
As it can be seen from the obtained electric field intensity distributions (Fig. 5), there is a great enhancement of the electric field at the corners of the nanoprisms (from tens to thousands times, depending on the nanoprisms shape and light polarization). If we compare the field enhancements at the corners of the trigonal and hexagonal nanoprisms, we shall see that it is about 10 times greater for the trigonal nanoprisms than for the hexagonal one. This implies the opportunity to exploit the trigonal nanoprisms for surface-enhanced spectroscopy.

To examine how the size of the nanoparticles influences the enhancement of the electric field, the same simulation was performed for trigonal and hexagonal Au nanoprisms of larger size (model 1 and model 2 in Fig. 3a) (Fig. 6). From these results it is evident that while the size of the Au nanoprisms increases from 25 to 100 nm (trigonal) and from 8 to 33 nm (hexagonal), the maximum field intensity enhancement increases approximately 10-fold.

In summary, the highest local electric field enhancement among the considered Au nanoprisms, up to 18000 times, was exhibited by the trigonal Au nanoprism with side dimension of 100 nm, when they are illuminated with light polarized normally to one of the triangular base sides. To achieve the maximum performance and reproducibility of such nanostructures for surface-enhanced spectroscopy and LSPR sensing applications, they should be prepared as highly ordered nanoparticle arrays with uniform shape and size.

5. Conclusions
In conclusion, it was confirmed for the studied trigonal and hexagonal Au nanoprisms deposited on the glass substrates that the LSPR spectra depend on the nanocrystals shape and size. Experimentally obtained and theoretically modeled LSPR positions differ due to the high spread of the Au nanoprisms shapes and sizes and aggregation of the nanocrystals. Local electric field intensity modeling under the LSPR conditions have demonstrated the promising potential of Au nanoprisms for surface-enhanced spectroscopy, with the highest enhancement provided by a 100 nm trigonal Au nanoprism on the glass substrate.

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